

# **Influence Of Dopants Fe<sup>2+</sup> And Yb<sup>3+</sup> On Structural And Optical Properties Of MnFe<sub>2</sub>O<sub>4</sub> Nanoparticles**

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## **Abstract**

*Co-precipitation is used to create cubic spinel manganese ferrite nanostructures that are both undoped and doped (Fe<sup>2+</sup>, Yb<sup>3+</sup>). X-ray diffraction (XRD), scanning electron microscopy (SEM), energy dispersive X-ray spectroscopy (EDX), and UV-visible absorption spectroscopy were used to examine the structural and optical characteristics of the nanoparticles. Through XRD patterns, a dopant's impact on microstructural parameters like size, cation radii, bond length, and lattice parameters is clearly visible. SEM images demonstrate the development of very dense nanocrystal formations and spherical shape. EDS spectra provide additional evidence of the nanoparticles' elemental composition. All findings show that the dopant has a major impact on the optical and microstructure of nanocrystals. The band gap of undoped MnFe<sub>2</sub>O<sub>4</sub> is 1.2 eV, of Fe doped samples varies from 2.2 – 2.7 eV for x = 0.1 – 0.6, of Yb doped samples varies from 2.10 – 2.40 eV for x = 0.025 – 0.2.*

**Keywords :** MnFe<sub>2</sub>O<sub>4</sub>; XRD; Band gap; bond length

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## **I. Introduction**

Cubic spinel structure can be represented by the general formula AB<sub>2</sub>O<sub>4</sub>, where divalent metal ions (A) occupy the tetrahedral sites (where A<sup>2+</sup>= Ni<sup>2+</sup>, Zn<sup>2+</sup>, Mn<sup>2+</sup> etc.) and trivalent metal ions (B) occupy octahedral sites (where B<sup>3+</sup>= Fe<sup>3+</sup>, La<sup>3+</sup>, Al<sup>3+</sup>, etc.)<sup>1-3</sup>. The scientific world has paid close attention to spinel ferrites, a significant class of magnetic materials that combine exceptional electrical, optical and magnetic properties. These materials are used in a wide variety of products, including switches, magnetic recorders, transformers, microwave devices, and computer memories, sensor<sup>4-7</sup>. Investigations into optical characteristics of these materials indicate that the properties are influenced by parameters during synthesis, doping element, preparation time, sintering temperature etc.<sup>8-10</sup>. Compared to other ferrites, manganese ferrite has a higher level of biocompatibility, making it more suitable for use in biomedical applications such drug delivery and magnetic resonance imaging (MRI)<sup>11,12</sup>. A review of the literature reveals that a variety of physical and chemical methods, including ball milling, sol-gel, hydrothermal, auto combustion, thermal treatment, and the co-precipitation method, were used to synthesise pure and doped manganese ferrites<sup>13-15</sup>.

According to literature survey, the optical band gaps depend on the size and composition of the materials. Thus, the band gap of the materials can be tailored for a particular application by altering either their composition or size<sup>16,17</sup>. Amulthy et al used the microwave combustion method to create Mn<sub>1-x</sub>Co<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> (x = 0, 0.1, 0.2, 0.3, 0.4, and 0.5). The band gap of pure manganese ferrite was observed to be 1.76 eV and grew to 2.25 eV when the concentration of Co<sup>2+</sup> ions increased<sup>18</sup>. Apostolova et al investigated effect Co, Mg and Ni doping on band gap of manganese ferrite and reported that band gap can increase or decrease using different ion doping<sup>19</sup>. By introducing a dopant, Bhalla et al. reported adjusting the band gap of Bi-doped nickel zinc (NiZn) spinel ferrite nanoparticles<sup>20</sup>. According to Devi et al., the optical band gap of nanoferrites Mn<sub>1-x</sub>Ni<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> (x=0.1, 0.2, 0.3, 0.4, and 0.5) is affected by Ni concentration, and a decrease in band gap was noted with an increase in Ni concentration<sup>21</sup>. Band gaps of 2.33 eV and 2.16 eV were found by Iranmahesh et al. for MnFe<sub>2</sub>O<sub>4</sub> at pH values of 9 and 11, respectively<sup>22</sup>. According to Jose et al., the bandgaps for MnF (manganese ferrite), MnCoF, and MnCoMgF were 3.202 eV, 3.180 eV, and 3.239 eV, respectively<sup>23</sup>. The band gap of the sample shrinks when Co<sup>2+</sup> is substituted in MnF, suggesting the quantum confinement phenomena. The band gap energy value rises when both Co<sup>2+</sup> and Mg<sup>2+</sup> were substituted.

There are no systematic reports on the optical properties of Fe<sup>2+</sup> and Yb<sup>3+</sup> doped manganese ferrite. In order to observe the effect on the band gap, Fe<sup>2+</sup> and Yb<sup>3+</sup> have been added to the manganese ferrite (MnFe<sub>2</sub>O<sub>4</sub>) system in the current work. The co-precipitation method is a useful technique for producing nanoparticles with a consistent composition in a short amount of time with a high yield. As a result, it was used to create the nanoparticles. No additional heat treatment is necessary for the procedure that is being provided. So, a straightforward low temperature chemical synthesis process was used to produce the Fe<sup>2+</sup> and Yb<sup>3+</sup> doped manganese ferrite nanoparticles. The preparation of Fe<sup>2+</sup> and Yb<sup>3+</sup> doped manganese ferrites nanoparticles

(MnFe<sub>2</sub>O<sub>4</sub>) and examination of their structural, optical, and morphological characteristics utilising XRD, FESEM, EDX, and UV spectroscopy are the objectives of this work. Due to their high surface-to-volume ratio and controllable band gap, the produced magnetic nanoparticles may find use in photocatalysis, dye degradation, waste water treatment technology, etc.

## II. Experimental

### Material Synthesis

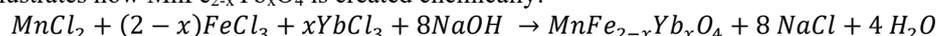
#### Fe<sup>2+</sup> doped manganese ferrite nanoparticles

The Mn<sub>1-x</sub>Fe<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> (x=0.1, 0.2, 0.4 and 0.6) nanoparticles were prepared using reactants manganese chloride, ferric chloride and NaOH as the precipitating agent by the technique of co-precipitation. MnCl<sub>2</sub>·4H<sub>2</sub>O solution (1M) and FeCl<sub>3</sub>·6H<sub>2</sub>O solution (2M) were combined and ultrasonically mixed for five minutes. The boiling solution of NaOH was quickly added after the two solutions had been combined and heated to 60 °C. A black precipitate with a strong magnetic response was immediately formed. For two hours at 85°C, the solution was stirred with a magnetic device. The precipitate was then magnetically separated, several times washed with de-ionized water to remove any remaining electrolytes, and then washed with ethanol once the reaction mixture cooled to room temperature. The sample was vacuum-dried in the last phase, after which it was ground into a powder using a mortar and pestle to create nanoparticles. The samples are labelled as MF0, MF0.1, MF0.2, MF0.4, and MF0.6. The chemical reaction for the synthesis of nanoparticles Mn<sub>1-x</sub>Fe<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub>:



#### Yb<sup>3+</sup> doped manganese ferrite nanoparticles

Using the co-precipitation technique, Yb-doped manganese ferrite nanoparticles were created. The measured amounts of FeCl<sub>3</sub> and YbCl<sub>3</sub>·H<sub>2</sub>O were dissolved in deionized water to form 2M solution and the stoichiometric amount of MnCl<sub>2</sub>·4H<sub>2</sub>O in deionized water (DI) was used to form 1.0M solution. The resultant solution (S1), obtained by mixing the above solutions, was heated to 60°C while being continuously stirred on a magnetic stirrer after being ultrasonically blended for 5 minutes at room temperature. In deionized water, a solution of NaOH was created (S2). This was heated before being combined with S1. Under magnetic stirring, the mixture was kept at 85°C for two hours. The final precipitate of synthesised nanoparticles was thoroughly washed in DI water multiple times before being rinsed with ethanol. The sample was vacuum-dried before being pulverised with a mortar and pestle to produce the MnFe<sub>2-x</sub>Yb<sub>x</sub>O<sub>4</sub> powder. MnFe<sub>2-x</sub>Yb<sub>x</sub>O<sub>4</sub> nanoparticles were produced using this process for x = 0.025, 0.075, 0.1, 0.15, and 0.2. The samples MnFe<sub>2-x</sub>Yb<sub>x</sub>O<sub>4</sub> (x = 0.025, 0.075, 0.1, 0.15, and 0.2) are labelled as MFYb0.025, MFYb0.075, MFYb0.1, MFYb0.15, and MFYb0.2. The following reaction illustrates how MnFe<sub>2-x</sub>Yb<sub>x</sub>O<sub>4</sub> is created chemically:



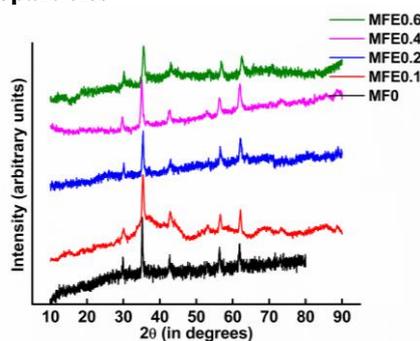
### Characterization

The powder XRD is captured using a Rigaku Ultima-IV X-ray diffractometer across a 2θ range of 20 to 80°. The crystal structure and phase purity of all the samples are investigated using a CuK<sub>α</sub> with a wavelength of 1.5406 Å and a Ni filter to remove CuK<sub>β</sub>. From XRD, microstructural parameters are also calculated. Using a Zeiss Gemini SEM500 equipped with energy-dispersive x-ray spectroscopy (EDS), the surface morphology, crystallite size, and microstructure of the materials were investigated. Using a Motras UV plus double beam UV-Vis spectrophotometer, absorption spectra between 200 and 900 nm were captured.

## III. Results and Discussion

### XRD studies

#### Fe<sup>2+</sup> doped Manganese ferrite nanoparticles



**Figure 1 : X-ray diffraction (XRD) pattern of MnFe<sub>x</sub>Fe<sub>2-x</sub>O<sub>4</sub> nanoparticles (where x = 0, 0.1, 0.2, 0.4, 0.6) : MF0, MFE0.1, MFE0.2, MFE0.4, and MFE0.4.**

The structural characteristics of Mn<sub>1-x</sub>Fe<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> nanoparticles, such as phase, crystallite size, and various bond lengths, are calculated from XRD of the samples as shown in Figure 1. XRD pattern of as-prepared Fe<sup>2+</sup> doped manganese ferrite nanoparticles amply demonstrates the existence of the pure ferrite phase. The peak was indexed to reflections (220), (311), (222), (400), (422), and (440), which supported the synthesis of a single-phase cubic spinel structure with space group Fd3m and absence of impurity phase, according to JCPDS Card No. 074-2403. As demonstrated in Fig. 1, the broadening of the (311) peak changes depending on the amount of Fe<sup>2+</sup> dopant present in the sample. Full width at half maximum (FWHM) is determined by the fitting of the most intense diffraction peak (311) using the pseudo-Voigt peak shape function to the most intense diffraction peak (311) in order to estimate the crystallite size (D) of the nanoparticles using Debye-Scherrer's formula<sup>24</sup>.

$$D = \frac{0.8\lambda}{\beta \cos(\theta)}$$

where  $\lambda$  is the wavelength of the X-ray wavelength used for diffraction, and  $\beta$  is the FWHM for the strongest peak at position ( $2\theta$ ). With more doping, the crystallite size shrinks. This might be because dopant Fe<sup>2+</sup> has smaller ionic radii than dopant Mn<sup>2+</sup>. Additionally, a change in the surface charge of nanoparticles may be the cause of this. The surface-to-volume ratio also varies due to the reduction in crystallite size, and this further alters the defect concentration. The hopping length, lattice parameter, and cation-oxygen bond length are all impacted by this shift in defect concentration. The various microstructural parameters can be calculated by taking into account the value of the oxygen positional parameter,  $u$  ( $\sim 0.381$ ), and the radii of oxygen ions,  $R_{O^{2-}} = 1.29 \text{ \AA}$ .

The ionic radii at the tetrahedral site ( $R_A$ ) and the octahedral site ( $R_B$ ), hopping length at the tetrahedral site ( $L_A$ ), hopping length at octahedral site ( $L_B$ ), cation-oxygen ion bond lengths in tetrahedral site ( $d_{Ax}$ ), cation-oxygen ion bond lengths in octahedral site ( $d_{Bx}$ ), tetrahedral distance ( $d_{AxE}$ ), octahedral shared distance ( $d_{BxE}$ ), and unshared distances ( $d_{BxEU}$ ) were calculated using the equations<sup>25</sup>

$$R_A = (u - 0.25)a\sqrt{3} - r_{O^{2-}}$$

$$R_B = (0.625 - u)a - r_{O^{2-}}$$

$$L_A = 0.25a\sqrt{3}$$

$$L_B = 0.25a\sqrt{2}$$

$$d_{Ax} = a\sqrt{3}(u - 0.25)$$

$$d_{Bx} = a \left[ \sqrt{3u^2 - 2.75u + 0.671875} \right]$$

$$d_{AxE} = a\sqrt{4u - 1}$$

$$d_{BxE} = a\sqrt{2 - 4u}$$

$$d_{BxEU} = a\sqrt{4u^2 - 3u + 0.6875}$$

$$d_{Ax} = a\sqrt{3}(u - 0.25)$$

$$d_{Bx} = a \left[ \sqrt{3u^2 - 2.75u + 0.671875} \right]$$

$$d_{AxE} = a\sqrt{4u - 1}$$

$$d_{BxE} = a\sqrt{2 - 4u}$$

$$d_{BxEU} = a\sqrt{4u^2 - 3u + 0.6875}$$

where  $a$  is the lattice parameter. Table 1 contains the calculated parameters. The table shows that all the parameters exhibit a reduction when the concentration of Fe<sup>2+</sup> dopant increases. This Table further supports that Fe<sup>2+</sup> dopant is successfully introduced into the lattice.

**Table 1 Microstructural parameters obtained from XRD**

sample	a (Å)	L <sub>a</sub> (Å)	L <sub>b</sub> (Å)	d <sub>AX</sub> (Å)	d <sub>BX</sub> (Å)	d <sub>AXE</sub> (Å)	d <sub>BxE</sub> (Å)	d <sub>BxEU</sub> (Å)	Crystallite size nm
MFE0	8.462	3.664	2.992	1.920	2.066	6.125	5.838	2.993	29.8
MFE0.1	8.413	3.643	2.974	1.909	2.054	6.090	5.804	2.976	12.6
MFE0.2	8.391	3.633	2.967	1.904	2.049	6.074	5.789	2.968	20
MFE0.4	8.382	3.630	2.963	1.902	2.046	6.068	5.783	2.965	15.8
MFE0.6	8.365	3.622	2.957	1.898	2.042	6.055	5.771	2.959	10.9

**Yb<sup>3+</sup> doped Manganese ferrite nanoparticles**

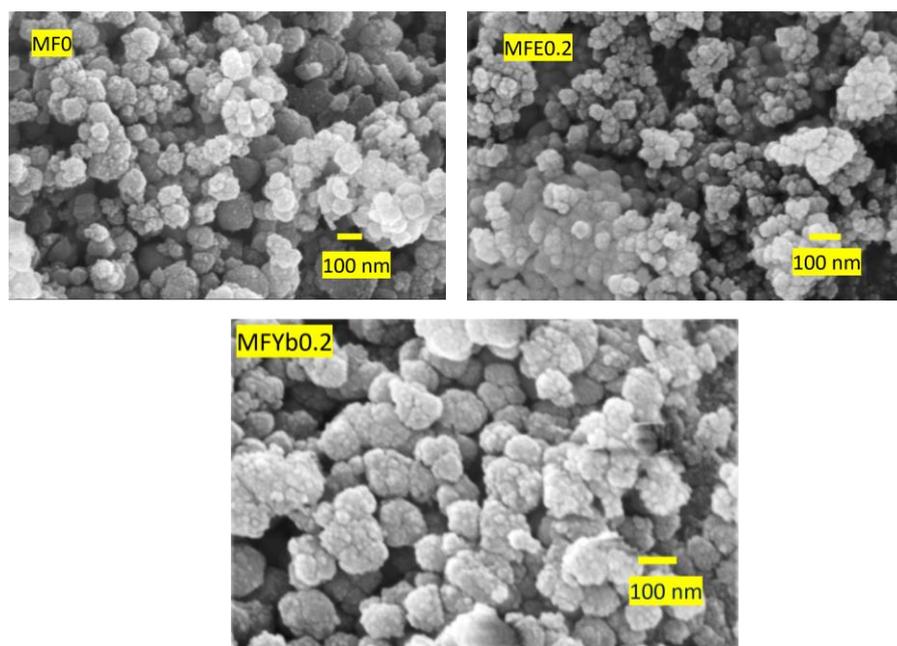
The face-centered cubic spinel structure of the synthesised Yb<sup>3+</sup> doped nanoparticles with high crystallinity was revealed by XRD measurements<sup>24</sup>. The peaks of Bragg diffraction and the XRD pattern of JCPDS card number 74-2403 are in agreement with the Fd-3m space group. XRD peaks were seen for phases (220), (331), (400), (333), and (440)<sup>24</sup>. No peaks are present because of any additional impurities or any other secondary phases. Because there were no further phases in the current work, it can be concluded that Yb<sup>3+</sup> ions are soluble in the host material and that the synthesis procedure was successful in incorporating Yb<sup>3+</sup> ions into the manganese ferrite matrix despite the large differences in ionic radii of Mn<sup>2+</sup> and Yb<sup>3+</sup> ions. From the sharp and XRD peaks, it can be inferred that the samples have good crystallinity. All of the XRD patterns of the doped samples showed the peak shifting which further supports that the Yb<sup>3+</sup> ions are incorporated in MnFe<sub>2</sub>O<sub>4</sub>.

**Table 2: The values of crystallite size (cs), lattice parameter (a), and density (ρ) of MnYb<sub>x</sub>Fe<sub>2-x</sub>O<sub>4</sub> nanoparticles**

Sample	cs (in nm)	a (in Å)	P (in g/cc <sup>3</sup> )
MF0	29.8	8.455	5.07
MFYb0.025	24.8	8.443	5.16
MFYb0.075	25.8	8.462	5.25
MFYb0.1	28.9	8.469	5.29
MFYb0.15	29.8	8.451	5.59
MFYb0.2	34.7	8.437	5.49

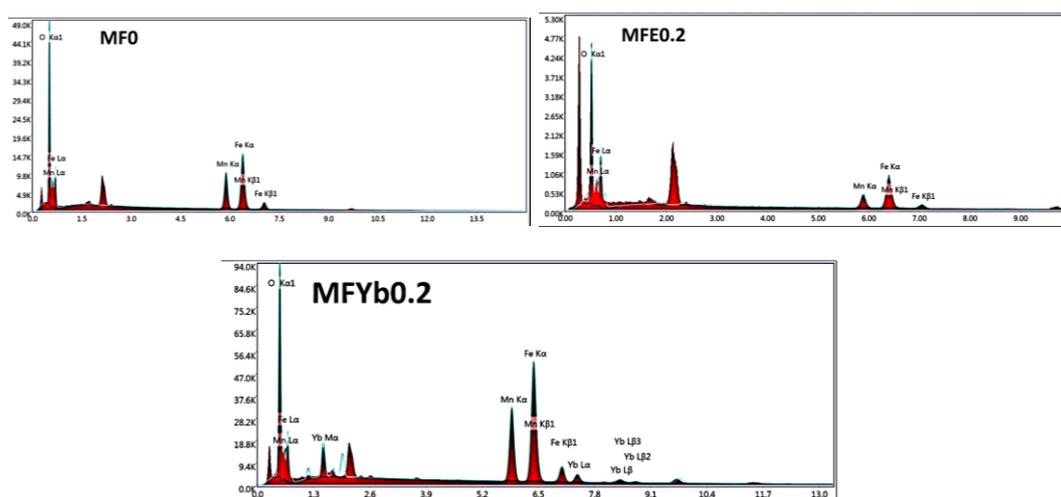
As ionic radius of Yb<sup>3+</sup> is more than the ionic radius of Fe<sup>2+</sup> ions, crystallite size of MnFe<sub>2-x</sub>Yb<sub>x</sub>O<sub>4</sub> nanoparticles is more than the crystallite size of Mn<sub>1-x</sub>Fe<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> nanoparticles (Table 2).

Though ionic radius of Yb<sup>3+</sup> ion (0.86Å) is more than the ionic radius of the Fe<sup>3+</sup> ion (0.63 Å), but it is observed that the substitution of Yb<sup>3+</sup> ions result in the crystallite size (cs) in the range of 24 – 28.9 nm, which is smaller than cs of undoped sample, for lower values of dopant concentration x (x ≤ 0.1). Such variations were also observed in the literature<sup>25,26</sup>. The smaller cs can be attributed to differences in electronic configurations, valencies, and ionic radii between rare earth ions and Fe<sup>3+</sup> ions [34]. The bond energy of Yb<sup>3+</sup>-O<sup>2-</sup> is greater as compared to that of Fe<sup>3+</sup> - O<sup>2-</sup>, thus an extra amount of energy is required to incorporate Yb<sup>3+</sup> ions into MnFe<sub>2</sub>O<sub>4</sub> to form Yb<sup>3+</sup>-O<sup>2-</sup><sup>27</sup>. This extra energy hinders the process of crystallization and hence the growth of grains in MnYb<sub>x</sub>Fe<sub>2-x</sub>O<sub>4</sub><sup>28</sup>. The substitution of Yb<sup>3+</sup> ions in spinel structure also leads to iron vacancies and distorts the tetrahedral and octahedral symmetry, affecting crystallite size lattice parameters. Besides this, some Yb<sup>3+</sup> ions may reside at the grain boundaries, which hinders grain growth and exerts pressure on the grain<sup>29</sup>. However, crystallite size for doped samples (for x = 0.15 and 0.2) is more than the undoped sample for higher values of x which could be attributed to the higher ionic radius of Yb<sup>3+</sup> as compared to that for Fe<sup>3+</sup>. In the present work, crystallite size is observed in the range 29.8nm – 34.7nm. Alonizan et al reported decrease in crystallite size (53.34nm – 28.77nm) with increase in concentration of dopant<sup>30</sup>.



**Figure 2** FE-SEM micrographs of (a)  $MnFe_2O_4$  nanoparticles (MF0), (b)  $Mn_{1-x}Fe_xFe_2O_4$  nanoparticles (MFE0.2), and (c)  $MnYb_xFe_{2-x}O_4$  nanoparticles (MFYb0.2).

The morphologies of undoped  $MnFe_2O_4$  nanoparticles (MF0),  $Mn_{1-x}Fe_xFe_2O_4$  nanoparticles ( $x=0.2$ ), and  $MnFe_{2-x}Yb_xO_4$  nanoparticles ( $x=0.2$ ), as representative sample, are depicted in FE-SEM micrographs in Figure 2. Sample MF0 sample had uneven flake-shaped nanoparticles, whereas samples that had been doped with  $Fe^{2+}$  and  $Yb^{3+}$  had isotropic spherical structures with homogenous uniform grain distribution. The average size of unsubstituted Mn ferrites is between 30 and 40 nm, whereas the average size of  $Fe^{2+}$  substituted  $MnFe_2O_4$  nanoparticles is between 10 and 25 nm and  $Yb^{3+}$  doped manganese ferrite nanoparticles have the average size between 30 – 50 nm. Figure 4 displays the EDX spectra of undoped  $MnFe_2O_4$  nanoparticles (MF0),  $Mn_{1-x}Fe_xFe_2O_4$  nanoparticles ( $x=0.2$ ), and  $MnFe_{2-x}Yb_xO_4$  nanoparticles ( $x=0.2$ ).



**Figure 3** : EDX spectrum of (a)  $MnFe_2O_4$  nanoparticles (MF0), (b)  $Mn_{1-x}Fe_xFe_2O_4$  nanoparticles (MFE0.2), and (c)  $MnYb_xFe_{2-x}O_4$  nanoparticles (MFYb0.2).

The samples included all the elements, including iron, oxygen, and manganese, according to the EDX spectrum. There were no signs of contaminants or other substances. Iron to oxygen (Fe:O) and metal ion composition ratios (Mn:Fe) / Mn:(Fe+Yb) were found to be consistent with the predicted composition ratio<sup>24</sup>. This demonstrates that the samples maintained the anticipated stoichiometry effectively. FESEM depicts agglomerated,

spherical-shaped nanoparticles. The EDS spectra of representative sample MnFe<sub>2</sub>O<sub>4</sub>, MnFe<sub>2-x</sub>Yb<sub>x</sub>O<sub>4</sub> (x = 0.2) indicate distinctive x-ray peaks of Mn, Fe and O atoms; and MnFe<sub>2-x</sub>Yb<sub>x</sub>O<sub>4</sub> indicates the distinctive x-ray peaks of Mn, Fe, Yb, and O atoms, but does not display peaks for any other element, proving the purity of the samples.

### UV-Vis Spectroscopy Studies

According to Tauc equation, the manganese ferrite nanoparticles' direct band gap ( $E_g$ ) is related to photon energy ( $h\nu$ )<sup>31</sup>

$$(\alpha h\nu)^2 = C (h\nu - E_g)$$

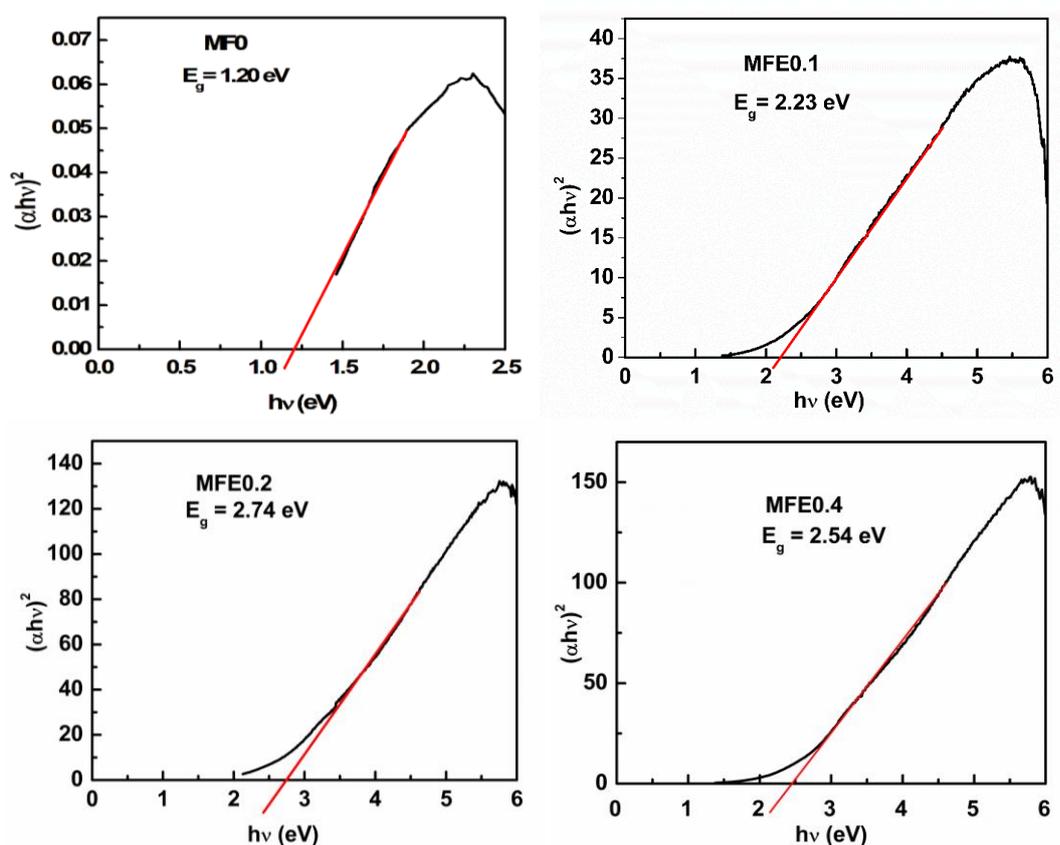
where  $\alpha$  is the absorption coefficient,  $h$  is Planck's constant,  $C$  is a constant of proportionality,  $E_g$  is the optical band gap of the material, and  $\nu$  is the frequency of radiation. The absorption coefficient  $\alpha$  was determined from the absorption data using the following equation

$$\alpha = 2.303 \left( \frac{A}{t} \right)$$

where  $A$  is the absorption and  $t$  is the thickness of the sample.

By extrapolating the linear section of the curve (Tauc plots of  $(\alpha h\nu)^2$  as a function of photon energy ( $h\nu$ ), to  $\alpha h\nu = 0$ , it is possible to calculate the band gap can be determined by extrapolating the linear region of the curve (Tauc plots of  $(\alpha h\nu)^2$  as a function of photon energy  $h\nu$ ) to  $\alpha h\nu = 0$ .

Tauc plots of  $(\alpha h\nu)^2$  as a function of photon energy  $h\nu$  is shown in Figure (4) for all the Mn<sub>1-x</sub>Fe<sub>x</sub> Fe<sub>2</sub>O<sub>4</sub> nanoparticles. The straight portion of the curve is extrapolated to determine the band gap.



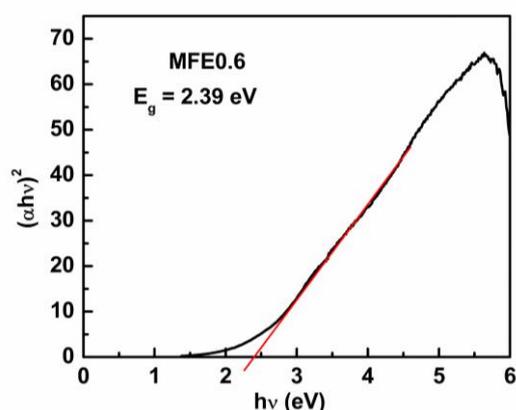


Figure 4 : Tauc plots of  $Mn_{1-x}Fe_xFe_2O_4$  nanoparticles : MF0, MFE0.1, MFE0.2, MFE0.4, and MFE0.6

The band gap of the samples fluctuates with doping and ranges from 1.20 eV to 2.74 eV, as seen in Figure 4. In the literature, such band gaps have been recorded<sup>32,21</sup>. The band gap is found to be at its maximum for the sample with  $x=0.2$ . Figure 5 and Table 1 show that the band gap variation with doping can be associated with crystallite size variation. The quantum size effects can be used to explain an increase in band gap with a decrease in crystallite size when doping concentration varies from  $x=0$  to 0.2. The decrease in the band gap with an increase in the crystallite size of the nanoparticles (for  $x=0.4$ , and 0.6) is attributed to surface and interface effects<sup>31, 22</sup>.

The band gap for doped  $MnFe_{2-x}Yb_xO_4$  nanoparticles falls in the range of 2.1 – 2.4 eV as shown in figure 5. Thus the band gap of doped samples is more than the band gap of undoped samples. The band gap increases with increase in  $x$ . The band gap varies inversely with crystallite size for the doped samples having  $x \leq 0.1$ . However, band gap varies directly with crystallite size for the doped samples having  $x = 0.15$  and 0.2. An increase in the band gap with a decrease in size of the nanoparticles is attributed to the quantum size effect and an increase in the band gap with an increase in the size of the nanoparticles is attributed to surface and interface effects.

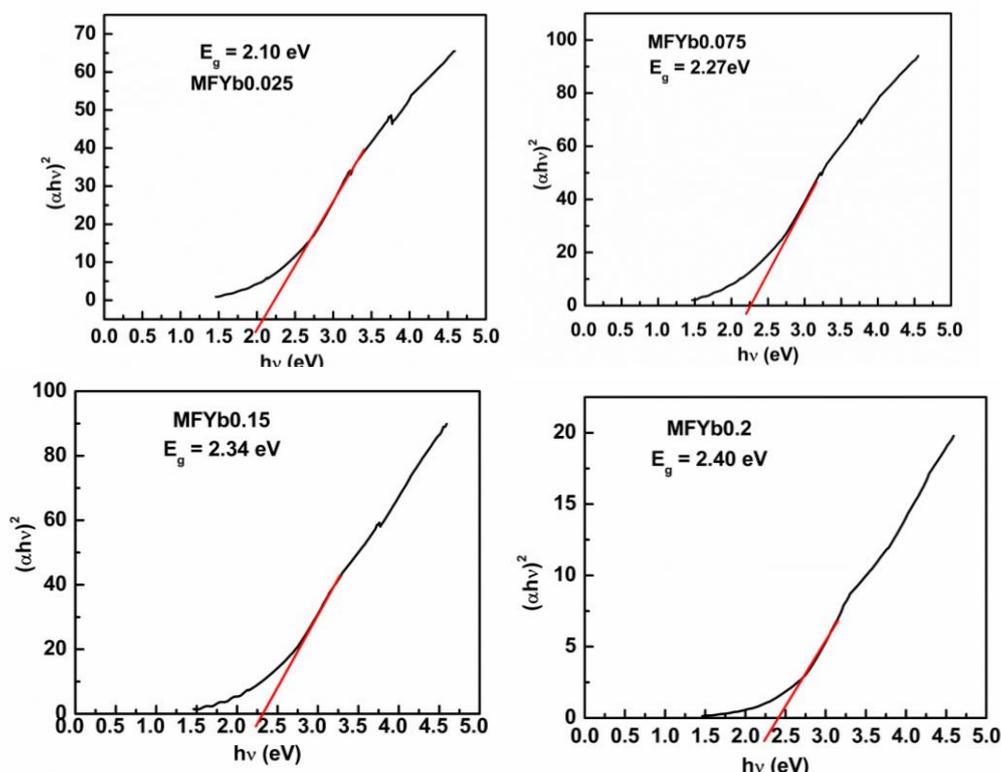


Figure 5 : Tauc plots of  $MnFe_{2-x}Yb_xO_4$  nanoparticles : MFYb0.025, MFYb0.075, MFYb0.15, and MFYb0.2

The band gap of Fe doped samples is higher than Yb<sup>3+</sup> doped samples. Although there is a correlation between  $E_g$  with  $cs$ ,  $E_g$  is dependent on multiple factors like crystallinity, synthesis method, process parameters, etc., hence it is difficult to quantify the trend.

It seems that additional sub-band-gap energy levels might be induced resulting in a smaller optical band gap when  $x > 0.2$  in Fe doped samples. However, band gap of Yb<sup>3+</sup> doped samples increase with increase in  $x$ , though crystallite size increases. The increase in the band gap with an increase in the crystallite size of the nanoparticles is attributed to surface and interface effects. The dopants can also be used to engineer band gaps for specific applications.

As photon energy of visible light ranges from 1.5 to 3 eV; Fe and Yb passes no threat to environment, thus doped manganese nanoparticles having a large surface-to-volume ratio with tuneable band gap can also be explored for applications in various fields such as solar cell technologies, opto-electronic devices, photovoltaic and photocatalytic degradation of pollutants.

#### IV Conclusions

Both undoped and doped (Fe<sup>2+</sup>, Yb<sup>3+</sup>) cubic spinel manganese ferrite nanostructures are produced by co-precipitation. As no additional heat treatment is required to produce nanoparticles, thus the Fe<sup>2+</sup> and Yb<sup>3+</sup> doped manganese ferrite nanoparticles were made using a simple low temperature chemical synthesis process. The structural and optical properties of the nanoparticles were investigated using X-ray diffraction (XRD), scanning electron microscopy (SEM), energy dispersive X-ray spectroscopy (EDX), and UV-visible absorption spectroscopy. A dopant's effect on microstructural factors like size, cation radii, bond length, and lattice parameters can be seen clearly by XRD patterns. FESEM photos show the growth of extremely dense nanocrystal formations and also their spherical form. EDS spectra offer additional proof of the elemental composition of the nanoparticles. The dopant significantly affects the optical and microstructure of nanocrystals, according to all available evidence. Undoped MnFe<sub>2</sub>O<sub>4</sub> has a band gap of 1.2 eV, Fe<sup>2+</sup> doped samples have band gap in the range of 2.2 to 2.7 eV for  $x = 0.1$  to 0.6, and Yb<sup>3+</sup> doped samples have a range of band gap 2.10 to 2.40 eV for  $x = 0.025$  to 0.2.

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